

Fabrication of transparent $\text{Cu}_x\text{Zn}_y\text{S}/\text{ZnS}$ heterojunction diodes by photochemical deposition

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Received ZZZ, revised ZZZ, accepted ZZZ

Published online ZZZ (Dates will be provided by the publisher.)

Keywords Please provide about four verbal keywords for your manuscript.

A $\text{Cu}_x\text{Zn}_y\text{S}/\text{ZnS}$ transparent pn heterostructure was successfully fabricated by the photochemical deposition (PCD) from aqueous solutions. In PCD, the substrate was immersed in the solution to a depth of about 2-3 mm from solution surface, and irradiated with the light of an ultra-high-pressure mercury arc-lamp through a lens. The growth solution of $\text{Cu}_x\text{Zn}_y\text{S}$ contained 5 mM CuSO_4 , 1 mM ZnSO_4 , 600 mM $\text{Na}_2\text{S}_2\text{O}_3$, and 3 mM Na_2SO_3 . The

growth solution of ZnS contained 1 mM ZnSO_4 , 600 mM $\text{Na}_2\text{S}_2\text{O}_3$, and 3 mM Na_2SO_3 . p-type conduction of $\text{Cu}_x\text{Zn}_y\text{S}$ and n-type conduction of ZnS were confirmed by the photoelectrochemical measurement. Both the films had high optical transmission (>70 %) in the visible range. The $\text{Cu}_x\text{Zn}_y\text{S}/\text{annealed-ZnS}$ heterostructure showed rectification properties and photo response to AM1.5 irradiation.

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1 Introduction Transparent or invisible electronics is one of the emerging major research areas. Extensive researches have been carried out to fabricate transparent devices, e.g., transparent transistors which can be used for display devices and transparent solar cells which can be installed as windows of buildings and houses. To realize transparent optoelectronic devices, both n-type and p-type transparent (wide-gap) semiconductor materials are indispensable. However, wide-gap semiconductors are mostly n-type without intentional doping because of native defects. The valence-control by doping has been attempted for various wide-gap materials, e.g., ZnO, ZnS, and SnO_2 , and transition from n-type to p-type has been observed [1-5]. However, those attempts have been still not successful enough to realize transparent electronics. When the valence-control is difficult, it is a common tactic to adopt a pn heterojunction structure: pn junctions can be fabricated using both an inherently n-type material and an inherently p-type material. So far, several p-type wide-gap materials have been found and investigated. Kawazoe et al. reported synthesis of CuAlO_2 , which has p-type conductivity and a band gap of about 3.5 eV [6]. Other Cu-based ternary oxides (SrCu_2O_2 [7], CuGaO_2 [8]) and sulfide (CuAlS_2 [9]) were found to have p-type conductivity and a band gap larger than 3 eV. There are also several reports on fabrica-

tion of transparent pn heterostructures based on p-type NiO [10-12]. It was found by our group that $\text{Cu}_x\text{Zn}_y\text{S}$ is p-type and transparent for visible light [13-15]. Since Cu_xS is a nonstoichiometric compound, i.e., the Cu/S ratio is not fixed, $\text{Cu}_x\text{Zn}_y\text{S}$ is also a nonstoichiometric material, i.e., the (Cu+Zn)/S ratio is not fixed. Therefore there are two independent variables x, y in its chemical formula. Recently, deposition of transparent and p-type $\text{Cu}_x\text{Zn}_y\text{S}$ films have also been reported by two other groups [16,17].

Cost and productivity of the deposition technique is another critical issue for actual applications. For display devices and solar cells, films need to be deposited in a large area at low cost. Moreover, the deposition temperature should be low for deposition on the conventional substrates such as glass and plastic films. Recently, chemical deposition techniques become increasingly popular since they satisfy the above requirements for the commercial production. Chemical bath deposition and electrochemical deposition (electrodeposition) have been extensively used to deposit various sulfides and oxides. Photochemical deposition (PCD) is another chemical technique of thin film deposition [18]. The substrate was immersed in the deposition solution, and irradiated with UV light. Then, owing to photochemical reactions in the solution, a compound is synthesized in the solution and deposited on the substrate.

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The PCD apparatus is simple and inexpensive, and films can be deposited at room temperature. PCD has been applied for various sulfides (e.g., CdS, ZnS, Cu_xS [18-20]) and related alloys (Cd_xZn_{1-x}S [21]). Synthesis of Cu_xZn_yS by PCD has been reported by our group [15].

In this work, we fabricate ZnS/Cu_xZn_yS heterostructures by PCD. So far, fabrication of heterostructures with ZnO has been reported [13,16], but ZnS/Cu_xZn_yS heterostructures have never been investigated. ZnS has a wider band gap than ZnO and thus has a wider transmission window. In addition, the interface structure may be simpler for ZnS/Cu_xZn_yS than for ZnO/Cu_xZn_yS, where a multinary transition layer can be formed because of mixing of both cations and anions. It should also be noted that the heterostructure fabrication by using PCD only is attempted for the first time in this work.

2 Experimental In PCD, the substrate was immersed in the solution to a depth of about 2-3 mm from the solution surface, and irradiated with the light of an ultra-high-pressure mercury arc-lamp through a lens. The light intensity was about 600 mW/cm². Indium-tin-oxide (ITO)-coated glass sheet was used for the substrate.

The growth solution of ZnS was an aqueous solution containing 1 mM ZnSO₄, 600 mM Na₂S₂O₃, and 3 mM Na₂SO₃ [19]. The pH of the solution was adjusted to be about 3.5 by adding H₂SO₄. Cu_xZn_yS films were deposited with the solution containing 5 mM CuSO₄, 1 mM ZnSO₄, 600 mM Na₂S₂O₃, and 3 mM Na₂SO₃. This solution composition is similar to that of ZnS deposition, but CuSO₄ is added. On the other hand, it is different from the solution conditions reported in the previous paper [15], and the reason for this difference will be discussed later.

In the PCD process of sulfides, S₂O₃²⁻ ions absorb UV light and release electrons and S atoms. Metal sulfide is formed by a reaction of metal ions with those electrons and S atoms. The reactions have been discussed in some more details in the previous paper [15].

For fabricating the heterostructure, the stacking order is an important factor, as discussed later. We fabricated heterostructures of both the order, i.e., ZnS/Cu_xZn_yS/ITO and Cu_xZn_yS/ZnS/ITO. For the latter structure, we annealed the ZnS layer in sulfur atmosphere. The sample with sulfur powder placed on the film surface was encapsulated between two glass slides and wrapped in aluminum foil to reduce loss of sulfur vapor [22], and annealed in vacuum at 400°C for 1 h. Indium electrodes of 1x1 mm² size were evaporated on the second layer surface.

Compositional analysis was performed using Auger electron spectroscopy (AES) employing a JEOL JAMP-9500F field emission microprobe. Argon ion etching was performed to sputter the film's surface. The atomic ratios were calculated using CuS, ZnS, Cu₂O and ZnO compounds as the standards. The film thickness was measured by an Accretech Surfcom-1400D profile meter. Optical characterization was performed using a JASCO U-570

spectrometer in reference to the ITO/glass substrate. To determine the conductivity type and evaluate photosensitivity of the films, photoelectrochemical (PEC) measurement was performed in a three-electrode electrochemical cell using a saturated calomel electrode (SCE) as the reference electrode and 100 mM Na₂S₂O₃ as the electrolyte. The sample was the film on the ITO glass substrate, and it was illuminated from the substrate side to evaluate the photoresponse using a Xe lamp with radiation power of 100mW/cm² as the light source under application of a ramp voltage. The incident light was turned off and on mechanically every 5 s. Current-voltage (I-V) characteristics of the heterostructure were measured in the dark and under AM1.5 irradiation from a solar simulator.

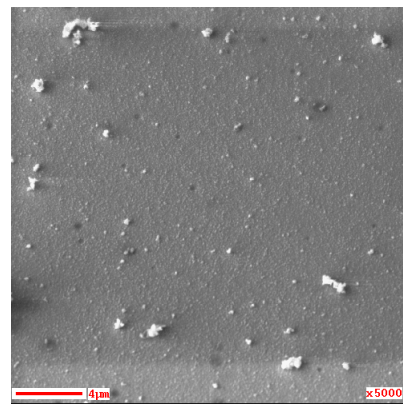


Figure 1 SEM image of the Cu_xZn_yS film.

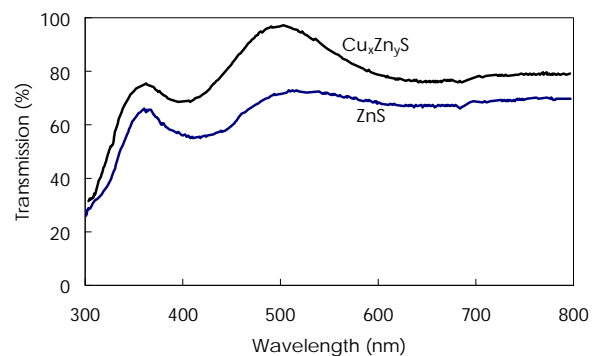


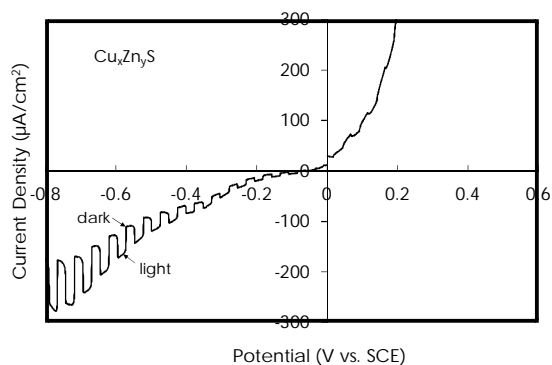
Figure 2 Optical transmission spectra for PCD-Cu_xZn_yS and ZnS.

3 Results and discussion

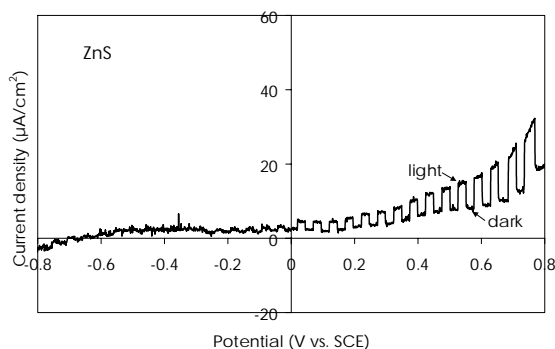
3.1 Cu_xZn_yS and ZnS films The thickness of the Cu_xZn_yS film is about 0.1 μm for a deposition time of 1 h. In the AES measurement, the Cu LMM signal appears around 915 eV, but it closely overlaps with a relatively weak peak of the Zn LMM signal. To observe the Cu signal, we subtracted the Zn LMM signal of ZnS from the Cu_xZn_yS spectrum after normalizing the peak intensity of the dominant Zn LMM peak at 995 eV. The elemental composition calculated from thus obtained AES data is Cu_{0.05}Zn_{0.61}S_{0.93}O_{0.07}. Thus the composition is Zn-rich, and

the metal/sulfur ratio is smaller than unity. The ZnS film is about 0.1 μm in thickness, and its composition obtained from the AES data is $\text{Zn}_{0.68}\text{S}_{0.88}\text{O}_{0.12}$. Figure 1 shows SEM photograph of the $\text{Cu}_x\text{Zn}_y\text{S}$ film surface. The film seems compact and flat.

Figure 2 shows the optical transmission spectra of the $\text{Cu}_x\text{Zn}_y\text{S}$ and ZnS thin films. For $\text{Cu}_x\text{Zn}_y\text{S}$, the transmission in the visible region was high, about 80%. For the ZnS film, the optical transmission in the visible range is fairly high but slightly smaller than that of $\text{Cu}_x\text{Zn}_y\text{S}$, probably because of inclusion of elemental Zn or S in the film. Thus, to improve the quality, the sulfur-annealing was carried out for the ZnS film as described in the next subsection. The band gap was estimated from the plot of $(\alpha h\nu)^2$ vs. $h\nu$, where α is the absorption coefficient and $h\nu$ the photon energy. The estimated band gap was about 3.7 eV for $\text{Cu}_x\text{Zn}_y\text{S}$. ZnS also has a comparable band gap. As reported in the previous paper, no X-ray diffraction peaks were observed for PCD $\text{Cu}_x\text{Zn}_y\text{S}$ and ZnS films, and thus they are amorphous or nano crystalline [15]. The band gap obtained here is slightly larger than the literature values for crystalline ZnS (3.5-3.6 eV), and this would be due to the amorphous nature of the films.



(a)



(b)

Figure 3 PEC measurement results for (a) $\text{Cu}_x\text{Zn}_y\text{S}$ and (b) ZnS.

Figure 3 shows the PEC measurement results for (a) $\text{Cu}_x\text{Zn}_y\text{S}$ and (b) ZnS. In the PEC measurement, current due to the minority carriers is significantly enhanced by the above-band gap illumination. Thus, the observed photocurrent is dominantly negative for a p-type semiconductor and positive for an n-type semiconductor. As shown in Fig.3(a), negative photocurrent was observed under the negative bias as for the $\text{Cu}_x\text{Zn}_y\text{S}$ film, and thus $\text{Cu}_x\text{Zn}_y\text{S}$ film is p-type. On the other hand, for ZnS, positive photocurrent was clearly observed under the positive bias, as shown in Fig. 3(b). Thus, n-type conductivity of ZnS was confirmed.

3.2 $\text{Cu}_x\text{Zn}_y\text{S}/\text{ZnS}$ heterostructures In a chemical solution deposition such as PCD, properties of the deposited film often depend on the substrate, because the initial nucleation is influenced by the substrate material. Moreover, during a heterostructure fabrication by a chemical process, the first layer may be dissolved into the deposition solution of the second layer. Then the properties of the interface and the second layer will be affected. Thus, the order of the deposition ($\text{ZnS}/\text{Cu}_x\text{Zn}_y\text{S}$ or $\text{Cu}_x\text{Zn}_y\text{S}/\text{ZnS}$) is an influential factor.

First, the $\text{ZnS}/\text{Cu}_x\text{Zn}_y\text{S}/\text{ITO}$ structures were fabricated. However, the ZnS layer on $\text{Cu}_x\text{Zn}_y\text{S}$ has a weak dark color, and the optical transmission in the visible range is only about 30 % for the heterostructure. Thus, the deposition of ZnS was affected by the underlying layer, or the $\text{Cu}_x\text{Zn}_y\text{S}$ is not stable in the ZnS deposition solution. The I-V characteristics of the $\text{ZnS}/\text{Cu}_x\text{Zn}_y\text{S}/\text{ITO}$ heterostructure are ohmic with a small resistance value (about 20 Ω).

Next, the $\text{Cu}_x\text{Zn}_y\text{S}/\text{ZnS}/\text{ITO}$ structures were fabricated. The appearance and thickness of the $\text{Cu}_x\text{Zn}_y\text{S}$ film on ZnS did not seem significantly different from those on the ITO substrate, and the heterostructure shows fairly high transmission in the visible region. Thus, ZnS seems stable in the $\text{Cu}_x\text{Zn}_y\text{S}$ deposition solution. However, for this structure also, the I-V characteristic is ohmic. Those poor electrical characteristics could be due to low quality of the ZnS films, because the ZnS films exhibited low optical transmission and small photocurrent compared with the $\text{Cu}_x\text{Zn}_y\text{S}$ film, as shown in the previous subsection. (We also fabricated $\text{ZnO}/\text{Cu}_x\text{Zn}_y\text{S}$ heterostructures and observed clear rectification properties. This also indicated that the $\text{Cu}_x\text{Zn}_y\text{S}$ films have quality high enough to act as a p-type layer of a diode. The results on $\text{ZnO}/\text{Cu}_x\text{Zn}_y\text{S}$ will be reported in another publication.)

Thus, to improve quality of the ZnS film, we annealed it in the sulfur ambient as described in the previous section. Figure 4 shows the optical transmission of the $\text{Cu}_x\text{Zn}_y\text{S}/\text{annealed-ZnS}$ heterostructure. The heterostructure appeared transparent: the transmittance was 70-80% in the visible region. Figure 5 shows the I-V characteristics of the $\text{Cu}_x\text{Zn}_y\text{S}/\text{annealed-ZnS}$ heterostructure. The weak rectification was observed, and thus crystallinity of the ZnS film was improved by the sulfur annealing. The current density is rather large because of poor rectification

properties. The current increased under the AM1.5 illumination, i.e., the heterostructure showed photoresponse. However, photovoltaic properties were not observed probably because of a large leakage current. For application for the transparent solar cell and LED, the interface quality needs to be improved.

As noted in the previous section, composition of the deposition solution for $\text{Cu}_x\text{Zn}_y\text{S}$ adopted in this work is different from that in the previous work [15]. In the previous work, the typical condition for obtaining transparent $\text{Cu}_x\text{Zn}_y\text{S}$ is as follows: 5 mM CuSO_4 , 25 mM ZnSO_4 , 300 mM $\text{Na}_2\text{S}_2\text{O}_3$. When $\text{Cu}_x\text{Zn}_y\text{S}$ was deposited on ZnS under this condition, the heterostructure was not transparent, i.e., the optical transmission in the visible range is 20-30 %. This could be explained as follows. The solution for $\text{Cu}_x\text{Zn}_y\text{S}$ in the previous work is significantly different from the ZnS deposition solution. Thus, ZnS was not stable and was dissolved or decomposed during the $\text{Cu}_x\text{Zn}_y\text{S}$ deposition, which affected the properties of the $\text{Cu}_x\text{Zn}_y\text{S}$ overlayer. On the other hand, the solution for $\text{Cu}_x\text{Zn}_y\text{S}$ adopted in this work is similar to that of the ZnS deposition. Therefore, dissolution or decomposition of ZnS was prevented, and the transparent $\text{Cu}_x\text{Zn}_y\text{S}$ film was deposited on ZnS.

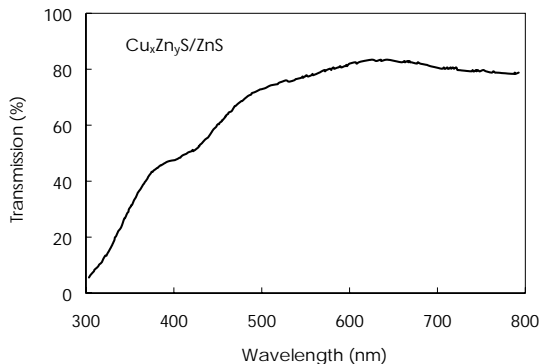


Figure 4 Optical transmission spectrum for the $\text{Cu}_x\text{Zn}_y\text{S}/\text{annealed-ZnS}$ heterostructure.

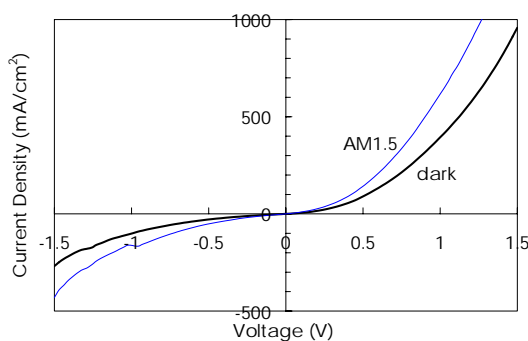


Figure 5 I-V characteristics in the dark and under AM1.5 illumination for the $\text{Cu}_x\text{Zn}_y\text{S}/\text{annealed-ZnS}$ heterostructure.

4 Summary A transparent pn heterostructure was successfully fabricated based on $\text{Cu}_x\text{Zn}_y\text{S}$ and ZnS films deposited by PCD. p-type conduction of $\text{Cu}_x\text{Zn}_y\text{S}$ and n-type conduction of ZnS were confirmed by the PEC measurement. Both the films had high optical transmission (>70 %) in the visible range. The $\text{Cu}_x\text{Zn}_y\text{S}/\text{annealed-ZnS}$ heterostructure showed rectifying I-V characteristics and photo response to AM1.5 irradiation.

Acknowledgements We would like to thank Dr. M. Kato for his useful discussion. This work was partly supported by a Grant-in-Aid for Scientific Research (C) from the Japan Society of Promotion of Science.

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